# Interfacial Electronics:

observing the buried physics of magnetic and superconducting nanostructures

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nm







### References

- Polarized Neutrons, W.G. Williams, Oxford (1988)
- Theory of Magnetic neutron and photon scattering, E. Balcar & S.W. Lovesey, Oxford (1988)
- Introduction to Thermal Neutron Scattering, G.L. Squires, Cambridge (1978)
- Elements of Modern X-Ray Physics, Als-Nielsen and McMorrow, Wiley & Sons (2001)
- Magnetism: from fundamentals to nanoscale dynamics, Stohr and Siegmann, Spinger (2006)
- www.ill.eu
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- www.esrf.eu
- www.diamond.ac.uk

# Spintronics and functional materials

The understanding of electronic behaviour in systems with reduced dimensionality and length scale is a centrel there of contemporary condensed matter physics. The unique capabilities of neutron scattoring make it an isleet mathed to study the atomic and molecular, shortical and magnetic structure of a wide class of materials. In this review we highlight recent studies where neutron bachniques have been applied to assurger instantials well look forward to the possibilities enabled by instrumentation on the ISIS Second Target Station.

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### Outline

#### Motivation

- The importance of interfaces
- Spatially resolving interfacial electronics
- Interfacial effects
  - A Magnetostructural Phase transition
  - Nanoscale Superconductivity

#### Outlook

Bright!





## Grand Challenges: Nanomagnetism



#### Nanoscale Electronic Phenomena Research



### Spintronics...





Fundamental understanding of electrons in reduced dimensions and lengthscale Cutting edge nanofabrication New physics Optimised functionality

Spintronics: Fundamentals and Applications. I. Žutić, J. Fabian, and S. Das Sarma, Rev. Mod. Phys. 76, 323-410 (2004)

Spintronics: a spin based electronics vision for the future S. A. Wolf, et al. Science 294, 1488 (2001);



# Quantum Well State



### Spintronics is new?

- " ... I have already communicated to the Royal Society a description of experiments by which I found that iron, when subjected to magnetic force, acquires an increase of resistance to the conduction of electricity across, the lines of magnetization..."
- William Thomson, "Effects of magnetization on the electrical conductivity of nickel and of iron" Proceedings of the Royal Society of London, Vol. 8, 1857, pp. 546550



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#### A history of storage/spintronics



- 1988
  - **Discovery of GMR**
- 1991
  - Spin Valve (IBM)
- 1994
  - MTI
- 1997
  - SV in a HDD
- 2004
  - Epitaxial MTJ
- 2005
  - **MTJ Read Heads**
- 2006
  - 4Mb MRAM





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Chappert, Fert, Nguyen van Dau Nat. Matl. 6, 813 (2007)



(2007)

#### **MRAM**

MagRAM Architecture Reading a bit Writing "T Writing "O Pit Lin, MTJ MagRAM promises - density of DRAM - opeed of SRAM - non-volatility 





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### Interference effects











Fresnel reflection 1815



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## Spin polarised Neutron Reflectivity

Measure the reflected neutrons as a function of their perpendicular momentum and spin eigenstate



- Can observe the magnitude and orientation of atomic magnetic moments in thin films and multilayer media.
- Probe length scale (<1nm to >1000nm): covers many aspects of thin film structure and magnetism
- Complementary to:
  - VSM/SQUID
  - MOKE average magnetisation over the sample thickness
  - SEMPA, Lorentz surface domain magnetisation
  - XMCD/XRMS-element specific
- PNR gives the microscopic *in-plane vector* magnetisation *depth* profile.



### Spin dependent cross-section

In-plane orientation of magnetisation obtainable from 4 spin dependent cross-sections

- Components of the magnetisation, m give rise to
- m||H: Non Spin Flip Scattering (NSF)
- m  $\perp$ H: Spin Flip Scattering (SF)
- Dynamical analysis gives absolute depth dependence profile

$$\left[\frac{-\hbar^2}{2m_n}\nabla^2 + V(r)\right]\Psi^{\uparrow,\downarrow} = E\Psi^{\uparrow,\downarrow}$$

NSF

 $b = b + p \sin \phi$ 

 $p_m \cos \phi = p_x$ 

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#### Small Angle Scattering

Grazing Incidence to give depth selectivity Difference gives the interference term

 $I^+(Q,\alpha) = \langle |F^{++}|^2 \rangle + \langle |F^{+-}|^2 \rangle$  $= F_N^2 + (F_M^2 - 2PF_NF_M)\sin^2\alpha$ 

$$I^{-}(Q,\alpha) = \langle |F^{--}|^2 \rangle + \langle |F^{-+}|^2 \rangle \\ = F_N^2 + (F_M^2 + 2P\epsilon F_N F_M) \sin^2 \alpha$$



#### Figure 10

2D-GISAXS pattern of an a-C:H/Au 8.2 at.% film using an imaging plate. The interference maximum (represented by a quasi-isotropic half ring) is related to the spatial correlation between isolated gold clusters embedded in the a-C:H matrix. The z-direction is perpendicular to the film surface (from Babonneau et al., 2001).



#### Figure 1

SANSPOL patterns in Fe<sub>3</sub>O<sub>4</sub> for neutron spins antiparallel (I) and parallel (I<sup>+</sup>) to the horizontal field. The arithmetic mean  $[(I) + (I^{+})]/2$  corresponds to the 2D pattern of non-polarised neutrons. The difference (I)-(I) yields the interference term [equation (1c)].



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#### A. Wiedenmann J. Appl. Cryst. (2000). 33, 428-432

### A Magnetostructural Phase Transition

#### FeRh epilayers

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PRB 82, 184418 (2010)

## Motivation

#### Tuneable Magnetostructural

#### nhaco transition



- FePt/FeRh system
  - Thiele et al. Appl. Phys. Lett. 82 2859 (2003)
- Control MTJ cell coercivity:
- 50Tb in<sup>2</sup>

- Magnetic refrigeration
  - ΔS ~ 15 J Kg<sup>-1</sup> K<sup>-1</sup>
- Ultra-fast switching
  - Ju et al. Phys. Rev. Lett. 93, 197403 (2004)





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# FeRh: Bulk Properties



- CsCl structure
  - a~2.99Å
  - $\alpha$ ' phase
- Isostructural
- 300K: Type G AF
- Fe: ~3.3 μ<sub>B</sub>
- Rh: no moment
- FM alignment within <111> planes
- AF alignment between <111> planes
- 350 K: AF  $\rightarrow$  FM
- Fe: ~3.1 μ<sub>B</sub>
- Rh: ~1 μ<sub>B</sub>



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Van Driel et al., J. Appl. Phys. 85, 1026 (1999) Shirane et al. Phys. Rev. 134, A1547 (1964)

D. Arena, BNL

# Sample Growth

- Base pressure: ~5×10<sup>-10</sup> torr
- Combination of Knudsen cells and electron beam hearths
- Substrate: MgO(001)
- Composition: 48% Fe, 52% Rh
- Deposition at 300°C; post-growth anneal at 800°C for 60 min.
- Cap at 300°C with either Au, Ta or MgO
- Nominal FeRh thickness: 22, 50, & 100nm







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### Structural

- Au/FeRh[50nm] (002)
- FeRh<001> | MgO<110>
- a= 2.998Å







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## Surface Diffraction

RT

- Surface: 2nm
- Bulk: 200nm





#### MgO[2]FeRh[50]/MgO



- I. Suzuki et al. J. Appl. Phys. 105, 07E501 (2009)
- Fan et al. prb 82 184418 (2010)



# Magnetic profile

- MgO -0.5%
- Clear evidence of RT FM
  - 📕 0.08μB
  - 📕 1.56μB @400K
- Rh rich surface
  - Reduced moment in FM phase
  - Reduced ordering temperature

7.0x10<sup>-6</sup> rength Density( $\mathring{A}_{-5}^{-2}$ ) endth Density( $\mathring{A}_{-5}^{-6}$ )  $4.0 \times 10^{-6}$   $4.0 \times 10^{-6}$   $3.0 \times 10^{-6}$ Scattering L 1.0x10<sup>-6</sup> 0.0





J. Van Driel et al., J. Appl. Phys. **85**, 1026 (1999)



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# Origin of ferromagnetism...

- Mix of  $\alpha$  and  $\gamma$  phases
  - Complex surface structure (Rh rich)
  - Favours reduction in AF-FM transition
- In-plane compression
  - Suppress the transition viz high pressure
  - Breaking of cubic symmetry
- Fe/Rh termination



- Suzuki *et al. japl 105, 07E501 (2009)*
- Lounis et al. prb 67 094432 (2003)
- Yamada et al. jac **415** 31 (2006)



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## Ultra-thin FeRh

PNR gives 230×10<sup>3</sup> Am<sup>-1</sup>
 Domain distribution studies

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- Suzuki et al. japl 105, 07E501 (2009)
- Lounis et al. prb **67** 094432 (2003)
- Yamada et al. jac **415** 31 (2006)

## Nanoscale Superconductivity



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# **DoE priorities**

#### Superconductivity Applications

 Developing HTS-based electric power equipment such as transmission and distribution cables and fault current limiters

#### Second-Generation Wire Development

Developing high-performance, low-cost, secondgeneration HTS wire at long lengths

#### Strategic Research

Supporting fundamental research activities to better understand relationships between the microstructure of HTS materials and their ability to carry large electric currents over long lengths

#### Applications

- SC Generators
- SC Energy Storage Systems
- SC Power Cables

#### Relevance of Neutron measurements

- Proximity effects
- Flux penetration
- Nonlocal effects
- FLL
- Control of T<sub>c</sub>
- Importance of Interfaces



http://www.oe.energy.gov





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#### Flux penetration in superconductors





R.J. Cubitt et al Phys. Rev. Lett. 91 047002 (2003)

- H<H<sub>c</sub> magnetic induction vanishes except within a penetration depth  $\lambda$
- H>H<sub>c</sub> destroys bulk s/c and magnetic flux penetrates sample
- $H_{c3}$ >H>  $H_{c2}$  surface s/c approximately equal to the coherence length,  $\varepsilon$
- Pb  $T_c = 7.2K$ ,  $H_c = 8000e$



#### Flux penetration in superconductors



FIG. 2. The spin-dependent reflectivities  $R^+$  and  $R^-$  measured in an applied magnetic field of  $6.0 \times 10^4$  A/m (750 Oe). The continuous lines are the reflectivities calculated for the two polarization states, with the same instrumental and surface parameters as in Fig. 1, and an exponential decay of magnetic induction with a penetration depth of 39 nm.

PHYSICAL REVIEW B

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Magnetic-induction profile in a type-I superconductor by polarized-neutron reflectom

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TABLE I. Literature results for the penetration depth of lead referred to absolute zero. Only the values for the actual penetration depths,  $\lambda(0)$ , are given even though in some techniques the measured quantity relates more naturally to the London penetration depth,  $\lambda_L(T)$ , at temperatures close to  $T_c$ .

Technique	$\lambda(0)/nm$	Ref.
Absolute surface impedance	~ 54	28
Magnetization of thin films	39±3	29
Perpendicular field transition	~44	30
Surface impedance	~48 <sup>a</sup>	31
Quantum interference in thin film	51-56	32
Field attenuation in thin film	45.3±8	33
Surface impedance in a field	~42	34
Absolute surface impedance	48±4	35
Inductance	~ 52.5	36
Polarized neutron reflectometry	$39\pm1$	This work

<sup>a</sup>Only  $\lambda_{I}(0)$  is given explicitly in Ref. 31, but the results for  $\lambda(T)$  obtained by a strong-coupling calculation are shown graphically in Fig. 1, and the value  $\lambda(0) \approx 48$  nm cited here has been taken from this graph.

> ino tiela dependence



 $H_c < H < H_{c3}$ 





#### TDGL





#### Magnetism and high temperature superconductivity

#### LETTERS

#### Magnetism at the interface between ferromagnetic and superconducting oxides

J. CHAKHALIAN<sup>1,2\*</sup>, J. W. FREELAND<sup>3</sup>, G. SRAJER<sup>3</sup>, J. STREMPFER<sup>1</sup>, G. KHALIULLIN<sup>1</sup>, J. C. CEZAR<sup>4</sup>, T. CHARLTON<sup>5</sup>, R. DALGLIESH<sup>5</sup>, C. BERNHARD<sup>1</sup>, G. CRISTIANI<sup>1</sup>, H.-U. HABERMEIER<sup>1</sup> AND B. KEIMER<sup>1</sup>

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4 nm

# Outlook I

- Interfacial phenomena offer the possibility of controlling magnetism
  - Magnetostructural transitions
  - Spin Accumulation
  - Dilute magnetic semiconductors
  - Hybrid magnetic/superconducting systems
  - Multiferroics ...
- Quantitative test of theory
- Interface Sensitive
- Dynamics/Kinetics
- Applicable to a wide class of nanoscale phenomena
  - Proximity effect
  - Dead layers
  - Vortex Structures (Reflectometry & SANS)











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# **Outlook II**

Sensitive & powerful techniques to spatially resolve the magnetism and superconductivity in nanoscale systems



Hrpd

Muons

